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Poly(Lactic Acid)-Based Biomaterials: Synthesis, Modification and Applications

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1. Introduction

Social and economic development has driven considerable scientific and engineering efforts on the discovery, development, and utilization of polymers. Widespread reliance in everyday life on conventional polymeric materials such as polyolefins has resulted in serious pollution which cannot be resolved in a straightforward fashion. Sustainable development and a green economy both require brand new materials which can avoid the occurrence of these problems.

Poly(lactic acid) (PLA), an aliphatic polyester, has outstanding advantages over other polymers, and may thus be part of the solution. As early as the 1970's, PLA products have been approved by the US Food and Drug Administration (FDA) for direct contact with biological fluids. Four of its most attractive advantages are renewability, biocompatibility, processability, and energy saving (Rasal, 2010). First of all, PLA is derived from renewable and degradable resources such as corn and rice, which can help alleviate the energy crisis as well as reduce the dependence on fossil fuels of our society; PLA and its degradation products, namely H₂O and CO₂, are neither toxic nor carcinogenic to the human body, hence making it an excellent material for biomedical applications including sutures, clips, and drug delivery systems (DDS). Furthermore, PLA can be processed by film casting, extrusion, blow molding, and fiber spinning due to its greater thermal processability in comparison to other biomaterials such as poly(ethylene glycol) (PEG), poly(hydroxyalkanoates) (PHAs), and poly(ɛ-caprolactone) (PCL) (Rhim et al., 2006). These thermal properties contribute to the application of PLA in industry in fields such as textiles and food packaging. Last but not least, PLA production consumes 25-55% less fossil energy than petroleum-based polymers. Cargill Dow has even targeted a reduction in fossil energy consumption by more than 90% as compared to any of the petroleum-based polymers for the near future, which will surely also lead to significant reductions in air and water pollutant emissions. It is also noteworthy that the total amount of water required for PLA production is competitive with the best performing petroleum-based polymers. This energy-saving feature perfectly caters to the

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new concept of "low-carbon economy" which emerged recently in response to the global warming and energy crisis concerns, and makes investment in PLA a necessary and wise strategy in the future (Vink et al., 2003). Fig. 1 shows the cycle of PLA in nature.



Fig. 1. The cycle of PLA in nature.

While PLA can be considered an eco-friendly biomaterial with excellent properties, it also has many obvious drawbacks when confronted with requirements for certain applications: 1) Its degradation rate through hydrolysis of the backbone ester groups is too slow. This process sometimes takes several years, which can impede its biomedical and food packaging applications (Bergsma et al., 1995). 2) PLA is very brittle, with less than 10% elongation at break, thus it is not suitable for demanding mechanical performance applications unless it is suitably modified (Rasal & Hirt, 2009). 3) PLA is strongly hydrophobic and can elicit an inflammatory response from the tissues of living hosts, because of its low affinity with cells when it is used as a tissue engineering material. 4) Another limitation of PLA towards its wider industrial application is its limited gas barrier properties which prevent its complete access to industrial sectors such as packaging (Singh et al., 2003). Considering the disadvantages of PLA stated above and its high cost (another shortcoming of that material), it is not surprising that PLA has not received the attention it deserves. Nevertheless, researchers have examined different methods for the bulk or surface modification of PLA, the introduction of other components, or the control of its surface energy, surface charge and surface roughness, depending on the requirements of specific applications.

Previous reviews have examined different aspects of PLA chemistry and engineering. Thus Maharana et al. (Maharana et al., 2009) presented a review on the melt-solid polycondensation of lactic acid (LA). Gupta et al. (Gupta et al., 2007) presented an overview of the production of PLA fibers by various methods, along with correlations between the

structure and the properties of the fibers. Butterwick et al. (Butterwick et al., 2009) discussed the applications of PLA in Europe and the United States with respect to practitioner experiences and techniques to optimize the outcomes. Rasal et al. (Rasal et al., 2010) examined the chemical modification of PLA, while Graupner et al. (Graupner et al., 2009) assessed the production and the mechanical characteristics of composites prepared from PLA and renewable raw materials including cotton, hemp, kenaf, and man-made cellulose fibres (Lyocell) by compression molding.

In this chapter we will underline novel ideas or technologies introduced over the last 5-10 years, emphasizing some ambitious work which, even though it appears less successful than other mature methods, introduces concepts that may prove extremely positive in the near future. We will also attempt to foretell developmental trends on the basis of social demands and the progress achieved so far. More traditional topics including the synthesis, modification, and applications of PLA in biomedical field will be introduced mainly to provide a more comprehensive picture of PLA as a biomaterial.

2. Physical and chemical properties of PLA

L-lactic acid and D-lactic acid, the two isomers of lactic acid, are shown in Scheme 1. Pure Llactic acid or D-lactic acid, or mixtures of both components are needed for the synthesis of PLA.



Scheme 1. The stereoisomers of lactic acid.

The homopolymer of LA is a white powder at room temperature with T_g and T_m values of about 55°C and 175°C, respectively. High molecular weight PLA is a colorless, glossy, rigid thermoplastic material with properties similar to polystyrene. The two isomers of LA can produce four distinct materials: Poly(*D*-lactic acid) (PDLA), a crystalline material with a regular chain structure; poly(*L*-lactic acid) (PLLA), which is hemicrystalline, and likewise with a regular chain structure; poly(*D*,*L*-lactic acid) (PDLA) which is amorphous; and *meso*-PLA, obtained by the polymerization of *meso*-lactide. PDLA, PLLA and PDLLA are soluble in common solvents including benzene, chloroform, dioxane, etc. and degrade by simple hydrolysis of the ester bond even in the absence of a hydrolase. PLA has a degradation half-life in the environment ranging from 6 months to 2 years, depending on the size and shape of the article, its isomer ratio, and the temperature. The tensile properties of PLA can vary widely depending on whether it is annealed or oriented, or its degree of crystallinity (Garlotta et al., 2001). Some of the physical and chemical properties of PLA are summarized in Table 1.

| Properties | PDLA | PLLA | PDLLA |
|----------------------------------|---|--|------------|
| Solubility | All are soluble in benzene, chloroform, acetonitrile, | | |
| | tetrahydrofuran (THF), dioxane etc., but insoluble in | | |
| | ethanol, methanol, and aliphatic hydrocarbons | | |
| Crystalline structure | Crystalline | Hemicrystalline | Amorphous |
| Melting temperature (T_m) / °C | ~180 | ~180 | Variable |
| Glass transition temperature | 50-60 | 55-60 | Variable |
| $(T_g)/°C$ | | $ \rangle \rangle \rangle \rangle \rangle \rangle \rangle $ | |
| Decomposition temperature/°C | ~200 | ~200 | 185-200 |
| Elongation at break/ (%) | 20-30 | 20-30 | Variable |
| Breaking strength/ (g/d) | 4.0-5.0 | 5.0-6.0 | Variable |
| Half-life in 37°C normal saline | 4-6 months | 4-6 months | 2-3 months |

Table 1. Selected physical and chemical properties of PLA.

3. Synthesis of PLA

Two main synthetic methods are used to obtain PLA: Direct polycondensation (including solution polycondensation and melt polycondensation), and ring-opening polymerization (ROP).

3.1 Direct polymerization

Since the LA monomer has both -OH and -COOH groups, necessary for polymerization, the reaction can take place directly by self-condensation (Scheme 2):

n HO
$$-\frac{CH_3}{H}$$
 COOH $\xrightarrow{\text{polymerization}}_{\text{catalyst}}$ H $\left(-\frac{CH_3}{H} - \frac{CH_3}{COO} \right)_n$ H + (n - 1) H₂O
Lactic acid Poly(lactic acid)
Scheme 2. Direct polymerization.

Direct polymerization includes solution and melt polycondensation, depending on whether a solvent is used in the reaction to dissolve the PLA or not.

3.1.1 Solution polycondensation

In this case an organic solvent capable of dissolving the PLA without interfering with the reaction is added, and the mixture is refluxed with removal of the water generated in the polycondensation process, which is beneficial to achieve a high molecular weight. Many procedures yield PLA with a weight-average molecular weight (M_w) of over 200,000 by this method (Ohta et al., 1995; Ichikawa et al., 1995). This approach was developed by Carothers and is still used by Mitsui Chemicals. The resultant polymer can be coupled with isocyanates, epoxides or peroxides to produce a range of molecular weights (Lunt et al.,

1998). The reaction proceeds smoothly, however solution polymerization suffers from certain disadvantages such as being susceptible to impurities from the solvent and various side reactions including racemization and trans-esterification. It also consumes large volumes of organic solvents, which are potential pollutants to the environment.

Under optimized conditions, Ajioka et al. obtained PLA with $M_w > 300,000$ by this method (Ajioka et al., 1995). Characterization data have shown that the glass transition temperatures (T_g) of PLA and polylactide synthesized by the conventional lactide process are essentially identical ($T_g = 58^{\circ}$ C and 59 °C, respectively), but PLA has a lower melting point ($T_m = 163^{\circ}$ C) than polylactide ($T_m = 178^{\circ}$ C). The mechanical properties of the two polymers are also very similar.

3.1.2 Melt polycondensation

In contrast to solution polycondensation, the melt polycondensation of monomers can proceed without any organic solvent, but only if the temperature of the reaction remains above the T_m of the polymer (Gao et al., 2002). Moon et al. discovered that high M_w PLLA $[M_w \ge 100,000]$ could be produced in this way in a relatively short reaction time (≤ 15 h) (Moon et al., 2000). This method can lower the cost of the synthesis significantly due to the simplified procedure, but major problems still need to be solved before it can be applied industrially because of its sensitivity to reaction conditions (Maharana et al., 2009). Thus Moon et al. worked to develop a melt/solid polycondensation technique using a binary catalyst system (tin dichloride hydrate and *p*-toluenesulfonic acid) (Moon et al., 2001). Simply put, thermal oligocondensates of LA were first subjected to melt polycondensation to obtain a melt polycondensate, which was then subjected to solid state polycondensation at 105°C. As a consequence, the molecular weight of the PLA was as high as 600,000 after a short reaction time under optimized conditions.

In summary, these one-step polymerization processes are relatively economical and easy to control, but they are equilibrium reactions affected by numerous parameters such as the temperature, the reaction time, catalysts, pressure, and so on. These factors can strongly influence the molecular weight of the products obtained. Besides, the water generated in this process can cause high molecular weight PLA to break down at high reaction temperatures. Thus the polymer resulting from these reactions usually has an unsatisfactorily low molecular weight. Attention must be paid to three aspects of the reaction to obtain a high molecular weight, namely controlling the reaction kinetics, removing the water formed, and preventing the degradation of the PLA chains.

3.2 Ring-opening polymerization

Considering the drawbacks of direct polymerization, PLA is typically synthesized by ringopening polymerization (ROP) (Scheme 3), an important and effective method to manufacture high molecular weight PLA. This reaction requires strict purity of the lactide monomer, obtained by dimerization of the lactic acid monomer. PLA is obtained by using a catalyst with the monomer under vacuum or an inert atmosphere. By controlling the residence time and the temperatures in combination with the catalyst type and concentration, it is possible to control the ratio and sequence of *D*- and *L*-lactic acid (LA) units in the final polymer (Lunt et al., 1998). The polymerization mechanism involved can be ionic, coordination, or free-radical, depending on type of catalyst employed (Penczek et al., 2000). Most of the researchers are now exploring new and effective catalysts. Köhn et al. (Köhn et al., 2003) first reported that the ROP of D,L-lactide could be catalyzed by bis(trimethyl triazacyclohexane) praseodymium triflate [(Me₃TAC)₂ Pr(OTf)₃] (Cat), while Pr(OTf)₃ by itself had a poor catalytic activity. Cat was found to catalyze the polymerization of *D*,*L*-lactide in various solvents (THF, dichloromethane, ethyl acetate, and toluene) without any additional reagents. The optimal polymer yield (95%) and molecular weight (18,000) were obtained after 18 h at 170°C, with a ratio of [LA]:[Cat] = 1000. John et al. (John et al., 2007) produced one of the few reports on lactide polymerization with a Cu-based catalyst. ROP of L-lactide catalyzed by {2-[1-(2,6-diethylphenylimino)ethyl]phenoxy}2Cu(II) yielded the highest M_w (26.3 × 10³) with a monomer conversion of 57%. Two other copper {2-[1-(2,6-dimethylphenylimino)ethyl]phenoxy}₂Cu(II) complexes, and {2-[1-(2methylphenylimino)ethyl]phenoxy}2Cu(II), also catalyzed the reaction under solvent-free melt conditions (160°C) but produced polylactides of moderate molecular weights (M_w = 12.0×10^3 and 15.9×10^3 , respectively).



Scheme 3. Ring-opening polymerization.

Numerous studies have examined the influence of different factors such as the concentration and type of catalyst, monomer purity, and temperature on the polymerization of lactide. Special attention has been paid to the catalyst. Currently tin octoate is the most widely used catalyst for the ring-opening polymerization of lactides, but numerous novel efficient metalfree catalytic systems are emerging as valuable alternatives (Jérôme & Lecomte, 2008). The heavy metal-based catalysts are indeed very likely to contaminate the product, which complicates the purification of the PLA obtained and also limits the applications of PLA in the fields of food packaging and biomedicine. Some of the means developed to solve this problem will be addressed later.

3.3 New approaches

The inherent disadvantages of the traditional synthetic methodology have led some researchers to explore solutions such as the development of nontoxic catalysts, unusual polymerization conditions, or other polymerization pathways (Lassalle & Ferreira, 2008). To solve the potential pollution problems caused by heavy metal catalysts, many nontoxic catalysts derived from magnesium (Wu et al., 2005), calcium (Zhong et al., 2003), zinc (Sarazin et al., 2004), alkali metals (Chisholm et al., 2003), and aluminum (Nomura et al., 2002) have been developed for the ROP of lactides (Deng et al., 2000; Ejfler et al., 2005). For example, Chen et al. (Chen et al., 2007) tested a series of β -diketiminate zinc complexes as initiators for the ROP of lactide and they were all highly active, however the M_w attained was unsatisfactory. It is worth noting that a variety of rare earth derivatives are usually highly reactive, which entitles them to be very promising initiators for the ROP of lactide (Agarwal et al., 2000).

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With respect to unusual reaction conditions, supercritical CO₂ (scCO₂) technology has attracted much attention because this environmentally friendly, chemically inert, inexpensive, non-toxic, and nonflammable solvent can be substituted for organic solvents (Nalawade et al., 2006). Yoda et al. (Yoda et al., 2004) thus carried out the synthesis of PLLA from an *L*-lactic acid oligomer in scCO₂ with dicyclohexyldimethylcarbodiimide (DCC) as an esterification promoter and 4-dimethylaminopyridine (DMAP) as a catalyst. PLLA with a number-average molecular weight M_n reaching 13,500 was obtained in 95% yield after 24 h at 3500 psi and 80°C. The molecular weight distribution of the products was also narrower than for PLLA prepared by melt-solid phase polymerization under conventional conditions. Not only can scCO₂ be used as a medium to synthesize polymers, but it can also serve in the purification and processing of the polymer microparticles obtained (Kang et al., 2008).

The direct polycondensation of lactic acid has been considered to have a promising future due to its low cost; however it is hard to increase the molecule weight due to the difficulty in removing the water from the system under these conditions. One way to solve this problem is a chain-extension method, although the properties of the PLA obtained in this way can be somewhat affected by the procedure. Simply put, hydroxyl- or carboxyl-terminated low molecular weight PLA obtained by direct polymerization can be linked together through a chain extender, which is a bifunctional compound carrying highly reactive functional groups. Many achievements have been reported in this area, hexamethylene diisocyanate (HDI) being the most widely used chain extender for hydroxyl-terminated prepolymers since the work done by Woo and coworkers (Woo et al., 1995). Finding new and satisfactory chain extenders will remain a major goal in the near future, since HDI is toxic and subject to side reaction in this process.

In addition, LA-polymerizing enzymes functioning in replacement of metal catalysts should enable the biosynthesis of PLA, even though it is enormously challenging both in terms of research and industrial implementation. The best solution could be the development of a PLA-producing microorganism, but this has not been reported so far. Taguchi et al. (Taguchi et al. 2008) have nonetheless obtained encouraging results by developing a recombinant Escherichia coli strain allowing the synthesis of LA-based polyesters by introducing the gene encoding polyhydroxyalkanoate (PHA) synthase. This is illustrated in Fig. 2. They thus achieved the one-step biosynthesis of a copolymer with 6 mol% of lactate and 94 mol% of 3-hydroxybutyrate units, having a molecular weight of 1.9 × 105. This extremely important result represents a milestone towards the biological synthesis of PLA and confirms that the work is moving in the right direction. At present, the LA fraction in the copolyesters has been enriched up to 96 mol% (Shozui et al., 2011), so the synthesis of homopolymers of LA represents a major goal. To that end, the current microbial cell factory ought to be improved with further evolved LA-polymerizing enzymes (LPE) and metabolic engineering-based optimization (Taguchi, 2010). Matsumura et al. (Matsumura et al., 1997) likewise reported the lipase PC-catalyzed polymerization of cyclic diester-D,L-lactide at a temperature of 80-130°C to yield poly(lactic acid) with molecular masses of up to 12,600. Other novel methods (e.g. metalfree catalysts, non-catalytic systems) are also under development (Zhong et al., 2003; Achmad et al., 2009). The advantages and disadvantages of the PLA synthesis methods mentioned above are summarized in Table 2.



Fig. 2. Mechanism for the bio-synthesis of LA polyester. In the bio-process, the LA monomer is converted into LA-CoA, which is recognized by the LA-polymerizing enzyme recruited from microbial PHA synthase (Tajima et al., 2009).

| Synthesis methods | Advantages | Disadvantages |
|-----------------------|----------------------------|---------------------------------|
| Solution | One-step, economical and | Impurities, side reactions, |
| polycondensation | easy to control | pollution, low molecular weight |
| | | PLA |
| Melt polycondensation | | High reaction temperature, |
| | | sensitivity to reaction |
| | | conditions, low molecular |
| | | weight PLA |
| Ring-opening | High molecular weight | Requires strict purity of the |
| polymerization | PLA | lactide monomer, related high |
| | | cost |
| New solutions | Efficient, non-toxic, no | Under development |
| (new catalysts, | pollution , high molecular | |
| polymerization | weight PLA, etc. | |
| conditions, etc) | | |
| Biosynthesis | One-step, efficient, non- | Under development |
| | toxic, no pollution, low | |
| | cost, etc. | |

Table 2. Comparison of PLA synthesis methods.

4. Modification of PLA

The major drawbacks of PLA limiting its applications are its poor chemical modifiability and mechanical ductility, slow degradation profile, and poor hydrophilicity. In order to be suitable for specific biomedical applications, the PLA has been modified mainly concerning two aspects: Bulk properties and surface chemistry. To achieve this, both chemical modification and physical modification have been tried, involving the incorporation of functional monomers with different molecular architectures and compositions, the tuning of crystallinity and processibility via blending and plasticization, etc., which are described in the following sections.

4.1 Bulk modification

Biomaterials must possess bulk properties, particularly hydrophilic and mechanical properties, meeting special requirements. Critical factors affecting these characteristics include chemical additives, composition, and morphological structure. At present considerable research focuses on a variety of hydrolytic groups, controlling the flexibility and crystallinity of the molecular chains, and the presence of hydrophilic groups.

4.1.1 Physical modification

Blending, plasticization, and composition variations belong to this category.

Blending

Polymer blending is an effective, simple, and versatile method to develop new materials with tailored properties without synthesizing new polymers (Peesan et al., 2005). The properties of different polymers (biodegradable and non-biodegradable) can be combined by blending with PLA, or even new properties can arise in the products due to interactions between the components. Biodegradable components blended with PLA include poly(ethylene glycol) (PEG), poly(β -hydroxybutyrate) (PHB), poly(ϵ -caprolactone) (PCL), poly(butylene adipate-*co*-terephthalate) (PBAT), chitosan, and starch (Sheth et al., 1997). While blends of PLA and non-biodegradable polymers have not been as extensively studied, low-density polyethylene (LDPE), poly(vinyl acetate) (PVA), and polypropylene (PP) have been examined. Reddy et al. (Reddy et al., 2008) found that PLA in blends obtained from five ratios of PLA/PP had substantially better resistance to biodegradation and hydrolysis, and improved dyeability with dispersed dyes. However most of these blends are immiscible (phase-separated) and display poor mechanical properties due to low interfacial adhesion between the polymer phases.

To improve the processing and mechanical properties of PLA without sacrificing its degradability and biocompatibility, Xu et al. (Xu et al., 2009) blended PLA with a new degradable thermoplastic derived from konjac glucomannan (TKGM), synthesized by graft copolymerization of vinyl acetate and methyl acrylate onto konjac glucomannan (KGM). Dynamic mechanical analysis (DMA) and scanning electron microscopy (SEM) measurements showed that the PLA/TKGM system was miscible due to specific interactions between PLA and TKGM. This led to a maximum elongation at break of 520% for the blend (20/80), as compared to 14% for neat PLA. The impact strength also increased from 11.9 kJ/m² for neat PLA to 26.9 kJ/m² for the 20/80 blend. The synthesis of new polymers, biodegradable or non-biodegradable, to be compatibly blended with PLA, will represent a major task in the future.

Plasticization

PLA is a glassy polymer with poor elongation at break (typically less than 10%). The modification of PLA with different biodegradable and non-biodegradable plasticizers,

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having a low molecular weight but a high boiling point and a low volatility, has been explored as a mean to lower the T_g and increase the ductility and softness of PLA. This has been achieved by varying the molecular weight, the polarity and functional groups of the plasticizers. Biocompatible molecules such as oligomeric lactic acid, oligomeric citrate ester, oligomeric PEG, and glycerol are all plasticizers of choice for PLA (Martin & Averous, 2001; Ljungberg et al., 2005). Ljungberg et al. (Ljungberg & Wesslén, 2002) have blended PLA with five plasticizers (triacetine, tributyl citrate, triethyl citrate, acetyl tributyl citrate, and acetyl triethyl citrate) and found that triacetine and tributyl citrate were more effective as plasticizers than the others to obtain a significant decrease in T_g for PLA.

Wang et al. (Wang et al., 2009) found that diisononyl cyclohexane-1,2-dicarboxylate (DINCH), a new plasticizer obtained by the hydrogenation of the benzene ring of ophthalates, had limited compatibility with PLA when compared with tributyl citrate ester (TBC). PLA samples plasticized with 10 and 20 phr DINCH gave a constant T_g of 50°C. They were stiff materials displaying elevated values of elongation at break (129% and 200%, respectively) and impact strength (41.1 MPa and 30.1 MPa, respectively). On the other hand, TBC significantly decreased the T_g and increased the crystallinity of PLA, the PLA/TBC (20 phr TBC) blend being a soft material with a T_g of 24°C. Results from thermogravimetric and thermal analysis also indicated that PLA plasticized with DINCH had good mechanical properties and excellent water resistance (as reflected in time-dependent weight loss data in phosphate buffer) and aging resistance (characterized by the mechanical and thermal properties of specimens exposed to ambient conditions for 4 months).

Composition

Fibers can serve as fillers in the formation of PLA composites processable by compression or injection molding, to enhance the thermal stability, the hydrolysis resistance, or the mechanical properties of PLA. Several investigations on PLA composites prepared from natural and modified cellulose fibers have shown that their mechanical properties scale with the mass fraction of added fibers (Wan et al., 2001; Mathew et al., 2005). Optimization of the natural fiber-reinforced PLA composites, in terms of mechanical and other properties, is critical to minimize their cost, tailor their biodegradability, and broaden their areas of application. Inorganic fillers can also contribute to property modification. Table 3 provides a comparison of some of the organic and inorganic materials tested as PLA fillers.

Graupner et al. (Graupner et al., 2009) prepared composites from different types of natural fibers (cotton, hemp, kenaf) and modified cellulose fibers (Lyocell), with a fiber mass fraction of 40%, by compression molding. The mechanical properties of these composites are summarized in Table 4. Tomé et al. (Tomé et al., 2011) prepared composites from PLA and acetylated bacterial cellulose by mechanical compounding. The composites displayed significant increases in both elastic and Young moduli, as well as in tensile strength (increments of about 100, 40, and 25%, respectively, as compared with neat PLA) at 6% filler loading. Some surface modifiers can enhance adhesion between the fibers and the PLA matrix. For example, 3-aminopropyltriethoxysilane (APS) hydrolyzes in water or solvents to produce silanol groups that are capable of bonding to -OH groups on the kenaf fiber surface (Huda et al., 2008). The -NH₂ groups from APS can also bond with -CO₂- sites formed on the PLA surface by treatment with a sodium hydroxide solution. Thus APS effectively functions as a coupling agent. Yang et al. (Yang et al., 2011) produced a composite from PLA and microcrystalline cellulose modified by *L*-lactic acid. The tensile strength and the elongation at break of the composite were higher than for neat PLA. The surface modification of the cellulose substrates was considered a key element of the mechanical reinforcement.

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| Туре | Filler | Result | Reference |
|-----------|--|--|-------------------------|
| Organic | Jute | Tensile stress and modulus increase with fiber volume fraction | Khondkeret al., 2006 |
| | Flax fibers | Composite strength about 50% higher than for PP/flax composites | Oksman et al., 2003 |
| | Kenaf fibers | Greatly improved crystallization rate, tensile and storage moduli | Pan et al., 2007 |
| | Bamboo fibers | Increased bending strength and improved thermal properties | Tokoro et al., 2008 |
| | Silkworm silk fibers | Good wettability, increased elasticity modulus and ductility | Cheung et al., 2008 |
| | Microcrystalline cellulose | Poor mechanical properties and adhesion; increased storage modulus | Mathew et al., 2005 |
| | LA-modified microcrystalline cellulose | Higher tensile strength and elongation at break than neat PLA | Yang et al., 2011 |
| | Acetylated bacterial cellulose | Considerable improvement in thermal and mechanical properties | Tomé et al., 2011 |
| Inorganic | Calcium metaphosphate | Narrow pore size distribution and high tensile strength | Jung et al., 2005 |
| | Calcium carbonate | No brittle fracture behavior and comparably high bending strength | Kasuga et al., 2003 |
| | Montmorillonite | Good affinity and improved thermal stability of the nanocomposites | Pluta et al., 2002 |
| | НАР | Improved elastic modulus and unchanged bending strength | Kasuga et al., 2001 |
| | Carbon nanotubes | Dramatic enhancement in thermal and mechanical properties | Wu & Liao, 2007 |
| | Nano/Micro-silica | Increased tensile strength, thermal stability, and hydrolysis resistance | Huang et al., 2009 |

Table 3. Organic and inorganic fillers for the preparation of PLA composites.

| \mathbf{P} | Tensile strength/ (N/mm²) | Young's modulus/ (N/mm²) | Elongation at break/ (%) | Charpy impact strength/ (kJ/mm²) |
|------------------|---------------------------------|--------------------------------|--------------------------------|--|
| Pure PLA sample | 30.1 | 3820.2 | 0.83 | 24.4 |
| Cottom-PLA | 41.2 | 4242.3 | 3.07 | 28.7 |
| Kenaf-PLA | 52.9 | 7138.6 | 1.05 | 9.0 |
| Hemp-PLA | 57.5 | 8064.2 | 1.24 | 9.5 |
| Lyocell-PLA | 81.8 | 6783.8 | 4.09 | 39.7 |
| Hemp/kenaf-PLA | 61.0 | 7763.8 | 1.22 | 11.8 |
| Hemp/Lyocell-PLA | 71.5 | 7034.9 | 1.65 | 24.7 |

Table 4. Mechanical properties of composites and a pure PLA sample (mean values; all the specimens were tested at 0°C; adapted in part from (Graupner et al., 2009).

Kim et al. (Kim et al., 2010) prepared a series of PLA/exfoliated graphite (EG) nanocomposites and confirmed that the graphite nanoplatelets could be dispersed homogeneously within the PLA matrix. Thermogravimetric analysis also showed that the thermal stability of the nanocomposites was improved with incremental amounts of EG up to 3 wt %. For example, the temperature corresponding to a 3% weight loss for a composite with 3.0 wt % EG increased by 14 degrees to ~364 °C vs. pure PLA. Additionally, the Young's modulus of the composites increased with their graphite content and their electrical resistivity was dramatically lowered. Poly(lactic acid)/hydroxyapatite (PLA/HAP) composite scaffolds processed by foaming with supercritical CO₂ were shown to be promising for bone replacement, because their mechanical characteristics closely matched the properties of bone in terms of viscoelasticity and anisotropy (Mathieu et al., 2006).

4.1.2 Chemical modification

The chemical modification of PLA has been achieved mainly through copolymerization and cross-linking.

Copolymerization

The carboxyl and hydroxyl groups of LA make it possible to copolymerize it with other monomers through polycondensation with lactone-type monomers such as ε -caprolactone, which generally leads to low molecular weight copolymers, or alternately through the ring-opening copolymerization of lactide with other cyclic monomers including glycolide, δ -valerolactone, and trimethylene carbonate, as well as with monomers like ethylene oxide (EO) to produce high molecular weight copolymers. The hydrophobicity and crystallinity of the copolymers can be increased for low to moderate commonly copolymerized with PLA to prepare copolymers on account that it is highly biocompatible, hydrophilic and non-toxic, non-immunogenic and non-antigenic (Metters et al., 2000). Such properties reduce protein adsorption and enhance resistance to bacterial and animal cell adhesion.

Block copolymers are composed of long sequences (blocks) of the same monomer unit, covalently bound to sequences of a different type. The blocks can be connected in a variety of ways. Fig. 3 shows examples of block copolymer structures. Diblock PLA-PEG copolymers and triblock PLA-PEG-PLA copolymers allow modulation of the biodegradation rate, the hydrophilicity, and the mechanical properties of the copolymers, while phase separation can be tailored with PLA-PEG multi-block copolymers of predetermined block lengths (Wang et al., 2005). Star- and dendrimer-like PLA-PEG copolymers have also been synthesized to lower the T_{g} , T_{nv} , and the crystallinity of the materials (Zhang et al., 2004).

Riley et al. (Riley et al., 2001) prepared a range of PLA-PEG copolymers incorporating a PEG block of constant molecular weight (Mn = 5,000) and varying PLA segment lengths (Mn = 2,000-110,000) by ROP of *D*,*L*-lactide catalyzed by stannous octoate; all the dispersions were stable under physiological conditions. In 2003, Li and Vert (Li & Vert, 2003) prepared series of diblock and triblock copolymers by ring-opening polymerization of *L*(*D*)-lactide from mono- and dihydroxyl PEO, using zinc metal as a catalyst under vacuum. The copolymers were semicrystalline, their composition and molar mass being determining factors affecting their solubility in water. Fu et al. (Fu et al., 2008) prepared series of LA-based polyurethanes modified by castor oil with controllable mechanical properties. In this work, hydroxyl-terminated prepolymers were synthesized by copolymerization of L-LA and 1, 4-butanediol.

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Fig. 3. Schematics of block copolymer structures: (a) diblock; (b) triblock; (c) alternating multiblock; (d); dendrimer-like copolymer; (e) star-like copolymer.

Cross-linking

Cross-linked PLA structures can be formed either by irradiation or through chemical reactions. Electron beam and γ -irradiation have been widely applied to cross-linking PLA in the presence of small amounts of cross-linking agents such as triallyl isocyanurate (TAIC) (Quynh et al., 2008; Phong et al., 2010). The thermal stability of PLA-based materials can be significantly improved in this way (Quynh et al., 2007). Quynh et al. (Quynh et al., 2009) obtained stereocomplexes by cross-linking blends of PLLA and low molecular weight PDLA. Alkaline hydrolysis and enzymatic degradation of the stereocomplex could be controlled by radiation cross-linking, because the alkaline solution as well as proteinase hardly attacked the cross-linked polymer network. Unfortunately, irradiation equipment is expensive and the PLA samples must be processed as thin plates to absorb enough energy from the radiation to initiate cross-linking reactions, which significantly limits its practical application.

Modified PLA with different gel fractions and cross-linking densities can also be obtained through chemical reactions between linking agents and the polymer chains without irradiation (Agrawal et al., 2010). Yang et al. (Yang et al., 2008) thus induced cross-linking via treatment of the PLA melt with small amounts of TAIC and dicumyl peroxide (DCP). The results obtained for samples with different gel fractions and cross-link densities showed that the cross-linking of PLA was initiated at low contents of either TAIC or DCP. The crystallinity of cross-linked PLA samples obtained with 0.5 wt% TAIC and 0.5 wt% DCP decreased from 32% for pure PLA to 24%. Significant increases in tensile modulus from 1.7 GPa to 1.9 GPa, and in tensile strength from 66 GPa to 75 GPa were also observed, and the thermal degradation initiation and completion temperatures were both increased relatively to neat PLA. Additional advantages of this method are that it requires neither extra purification steps nor specialized equipment, since the reaction is carried out in the molten state with only small amounts of cross-linking agent. It is thus economically very advantageous over irradiation, which requires expensive equipment. An increase in brittleness was nevertheless observed following the formation of highly cross-linked structures, which remains a problem to be solved.

4.2 Surface modification

The surface properties of materials play a key role in determining their applications. The presence of specific surface chemical functionalities, hydrophilicity, roughness, surface energy, and topography is crucial for biomedical applications of PLA and its interactions with biomacromolecules. Pure PLA causes a mild inflammatory response if it is implanted into human tissues. It is therefore important to design biomaterials with the required surface properties. The different surface modification strategies examined include physical methods, including surface coating, entrapment and plasma treatment, and chemical methods. Both types of approaches are reviewed.

4.2.1 Physical methods

Surface coating

This is one of the simplest surface modification methods and has been applied to various polymers, but particularly to PLA nanoparticles used for drug delivery. For instance, PEG coating delayed the phagocytosis of PLA nanoparticles and prolonged the circulation time of the nanoparticles in vivo (Gref et al., 1994). Unfortunately the PEG-coated PLA nanoparticles cannot provide specific targeting, which influences their delivery efficiency. One of the most promising alternatives to PEG in this respect is the use of polysaccharides. These materials provide steric protection to the nanoparticles against non-specific interactions with proteins and thereby insure particle stability in the blood circulation system (Ma et al., 2008). Additionally, ligands to achieve active targeting can be conjugated on the surface of these nanoparticles, because many reactive groups are available on the polysaccharides and their derivatives (Gu et al., 2007). Another option is coating of the surface with extracellular matrix (ECM) proteins such as fibronectin, laminin, vitronectin, and collagen, which are conducive to cell adhesion and can greatly improve biocompatibility as well (Lin et al., 2010).

Innovative work was accomplished by Cronin et al. (Cronin et al., 2004), who tested a PLLA fiber scaffold as a substrate for the differentiation of human skeletal muscle cells. Cell attachment (the number of cells attached to the films counted along the center, from one edge to the opposite edge of the film within the field of view) increased significantly on PLLA films coated with ECM gel, fibronectin, or laminin as compared to uncoated or gelatin-coated PLLA films. Myoblasts were able to differentiate into multinucleated myofibers on the ECM gel-coated PLLA fibers and expressed muscle markers such as myosin and α -actinin, as demonstrated by western blot and oligonucleotide microarray analysis.

Entrapment

The entrapment of modifying species (e.g. PEG, alginate, gelatin, etc.) can be achieved through reversible swelling of the PLA surface as illustrated in Fig. 4. This is a simple yet effective method for surface modification requiring no specific functional groups in the polymer chains, as the modifying molecules accumulate merely on the surface of the material without modifying its bulk properties (Lu et al., 2009). Additionally, entrapment can be used to generate different morphologies and thicknesses of 3D scaffolds, which cannot be achieved by other surface modification methods. Finally, entrapment allows the modification of the surface in a controlled fashion because various parameters (e.g. solvent ratio, gelatin concentration, immersion time, and chemical cross-linking) can be varied to tailor the process (Zhu et al., 2003).



Fig. 4. Schematic illustration of entrapment process.

PEG ($M_w = 18,500$) and poly(*L*-lysine) (PLL) ($M_w = 29,300$) have been trapped on PLA surfaces using 2,2,2-trifluoroethanol (TFE)/water as solvent/nonsolvent mixtures (Quirk et al., 2002). A new entrapment process has also been reported by Liu et al. (Liu et al., 2005), through chemical cross-linking of gelatin with 1-ethyl-3-(3-dimethylaminopropyl) carbodiimide (EDC) HCl and *N*-hydroxysuccinimide (NHS) (97%) in {2-[N-morpholino] ethanesulfonic acid} (MES) hydrate buffer, after the pretreated PLLA films were immersed in the gelatin solution for a set time. Results in comparison to the control scaffolds have shown that the surface hydrophilicity increased with the amount of entrapped gelatin and that cell attachment and proliferation, the deposition of collagen fibers, and other cell excretion (extracellular matrix, etc.) were also significantly improved.

Plasma treatment

Tests with plasma treatment were initiated in the 1960s and have been since then widely utilized to improve the hydrophilicity and cell affinity of PLA surfaces. The obvious advantages of plasma treatment as compared to other surface modification methods include its ability to control the surface structure, energy and charge, and to uniformly modify the surface without impacting bulk properties (Chu et al., 2002). Functional groups such as – NH₂, –COOH, and –OH, which are apt to form covalent bonds with other materials for further modification, are most frequently introduced by plasma treatment (Favia et al., 1998).

Liu et al. thus investigated the influence of the main operation parameters, namely the plasma power, the treatment duration (number of treatment cycles) and the electrode gap on a dielectric barrier discharge (DBD) plasma treatment of PLA films in terms of changes in surface wettability and chemistry (Liu et al., 2004). They further developed equations relating the surface properties (water contact angle and oxygen enrichment, as observed by XPS analysis) to these operational parameters. It was determined that the magnitude of the electrode gap played a dominant role in the treatment of PLA, and the observed wettability improvements were attributed to changes in both surface chemistry and microstructure. Chaiwong and coworkers (Chaiwong et al., 2010) investigated the influence of SF₆ plasma on the hydrophobicity and barrier properties of PLA. It was found that the SF₆ plasma enhanced the hydrophilicity and increased the water absorption time of PLA two-fold. Plasma treatment did not have any significant influence on the water vapor permeability of PLA, however, since the bulk structure controlling the transport properties are unaffected by the treatment. Other types of plasmas such as oxygen, helium, and nitrogen plasmas have also been investigated (Hirotsu et al., 2002).

While plasma treatment has been successfully applied to improving PLA wettability and cell affinity, its main disadvantage is that surface rearrangements caused by thermally activated macromolecular motions, to minimize its interfacial energy, can also influence the surface modification. Moreover, the potential influence of plasma on the degradation of PLA cannot be ignored.

4.2.2 Chemical modification

PLA does not carry reactive side-chain functional groups. Consequently, the first step of chemical modification is typically a simple surface hydrolysis (with an alkali) or an aminolysis treatment. The hydrophilic –COOH and –OH or reactive –NH₂ groups introduced by cleavage of the ester bonds can be used to bind bioactive molecules such as arginine-glycine-aspartic acid (RGD)-containing peptides, chitosan (CS), arginine and lysine, PEG, collagen, and so on to regulate cell adhesion or protein adsorption.

The synthetic RGD-containing peptides could be immobilized on PLA after treatment by hydrolysis or aminolysis (Stupack et al., 2001). Materials prepared by this method provide suitable recognition sites for cell adhesion receptors and biodegradation rates, making them suitable for various applications in fields such as tissue engineering and implant technology. It has also been determined that RGD-conjugated poly(lactic acid-*co*-lysine)(arginine-glycine-aspartic acid) nanoparticles (PLA-PLL-RGD NP) are non-toxic and bind more efficiently to human umbilical vein endothelial cells (HUVECs) as compared to bare PLA-PLL NP in vitro. Targeted imaging results obtained in vivo showed that PLA-PLL-RGD can selectively bind to BACP-37 breast cancer cells. Lieb et al. also demonstrated largely increased cell densities and cell proliferation on surfaces modified with RGD-anchored monoaminated poly(ethylene glycol)-*block*-poly(*D*,*L*-lactic acid) (H₂N-PEG-PLA), mediated through RGD-integrin interactions (Lieb et al., 2005).

Chitosan (CS) is a biopolymer displaying good biocompatibility, non-toxicity and biodegradability, produced by the alkaline *N*-deacetylation of chitin. The immobilization of this polymer on PLA has been accomplished by coating the surface with chitosan, modified with the photosensitive hetero-bifunctional cross-linking reagent 4-azidobenzoic acid, and irradiation with ultraviolet light to photolyze the azide groups and covalently link the two polymers (Zhu et al., 2002). The -OH and -NH₂ groups of chitosan provide further opportunities to introduce a wide range of functional groups on the surface. Thus CS molecules immobilized on PLA were modified with a heparin (Hp) solution to form a polyelectrolyte complex on the surface, which inhibited platelet adhesion and activation, and enhanced cell adhesion.

4.3 Outlook of PLA surface modification

The surface attributes of PLA can be tailored to enhance its hydrophilicity and biocompatiblity through various methods. Unfortunately, all these established methods for surface modification are inherently flawed to some extent. For example, a single plasma treatment can merely improve cell adhesion but cannot accelerate cell growth; non-covalent attachment of a functional material onto a PLA surface is not stable and permanent. An excellent method suggested to solve the second issue is the use of 1,6-hexanediamine for surface aminolysis, followed by conjugation with biocompatible macromolecules such as gelatin, chitosan, or collagen (Zhu et al., 2004). Hong et al. have

shown that chondrocyte cells could attach, proliferate, and spread on PLA microspheres coated with collagen in the same way as described above, in particular those having high collagen contents (Hong et al., 2005). It appears that the surface modification of PLA would be best achieved with a combination of distinct approaches, to benefit from the advantages of all the methods. Polysaccharide polyelectrolyte multilayers, including chitosan and dextran sulfate-stabilized silver nano-sized colloids developed by Yu et al. (Yu et al., 2007), were successfully deposited on an aminolyzed PLA membrane in a layerby-layer self-assembly manner. This seemingly easy process resulted in significant improvements in hydrophilicity, antibacterial activity, hemocompatibility, and cytocompatibility for the PLA membrane, thanks to the different attributes of -NH3+ (positive charge), chitosan (biocompatibility), and silver nanoparticles (antibacterial activity). The radiation-induced methods are emerging as powerful surface modification techniques, particularly when relying on PLA photoactivation to create reactive groups or moieties useful to graft specific chemical functionalities. The irradiation of PLA with UV (ozone can be generated from molecular oxygen irradiated with UV in this process), for example, is known to increase fiber adhesion to high surface energy components due to the introduction of photo-oxidized polar groups on the surface (Koo & Jang, 2008). Irradiation followed by grafting has also been used extensively to alter PLA surface characteristics, mainly due to the advantages it offers, namely a low operation cost, mild reaction conditions, selectivity to UV light, and the permanent surface chemistry changes induced (Ma et al., 2000).

5. Applications of PLA in the biomedical field

Due to its bioresorbability and biocompatibility in the human body, PLA has been employed to manufacture tissue engineering scaffolds, delivery system materials, or covering membranes, different bioabsorbable medical implants, as well as in dermatology and cosmetics.

5.1 Tissue engineering

Since the introduction of the concept in 1988, tissue engineering, a technique invented to reconstruct living tissues by associating the cells with biomaterials that provide a scaffold on which they can proliferate three-dimensionally and under physiological conditions, has emerged as a potential alternative to tissue or organ transplantation and has thus attracted great attention in science, engineering, and medicine. To meet the diverse needs of tissue engineering, scaffolds made from various materials have been tested in this field. Although certain metals are somewhat good choices for medical implants due to their superior mechanical properties, their lack of degradability in a biological environment makes them disadvantageous for scaffold applications (Liu & Ma, 2004). Inorganic/ceramic materials such as HAP or calcium phosphates, being studied for mineralized tissue engineering with good osteoconductivity, are also limited due to poor processability into porous structures (Ilan & Ladd, 2002). In contrast, polymers have great design flexibility because their composition and structure can be tailored to meet specific needs (Huang et al., 2007). Degradable polymers frequently used for tissue engineering applications are linear aliphatic polyesters such as PGA, PLA, and their copolymers (PLGA), which are fabricated into

scaffolds. These polymers are among the few synthetic polymers approved by the FDA for human clinical applications. The drawbacks of these polyesters include their hydrophobicity and lack of functional groups, which limits cell adhesion, an important factor when constructing polymeric scaffolds. Another drawback is their slow hydrolytic degradation (Iwata & Doi, 1998).

An ideal scaffold used for tissue engineering should possess the following properties: 1) Be biocompatible, so that the scaffold can be well integrated into host tissues without resulting in any immune response; 2) It should be porous with appropriate pore size, size distribution and mechanical function, to allow cell or tissue growth and the removal of metabolic waste; 3) It must be mechanically able to withstand local stress and maintain the pore structure for tissue regeneration; 4) Very importantly, the scaffold should be biodegradable (Ma, 2004). Synthetic scaffolds are considered important components of a successful tissue engineering strategy (Wang, 2007). Hybrid three dimentional porous scaffolds of synthetic and naturally derived biodegradable polymers are particularly promising because they combine the advantages of the two types of materials. They should maintain sufficient mechanical strength while providing specific cell-surface receptors during the tissue remodeling process that stimulate both in vitro and in vivo cell growth (Chen et al., 2002). PLA-based hybrid materials have been successfully tested clinically for that purpose so far, and tests on other tissues including bladder (Engelhardt et al., 2011), cartilage (John et al., 2003), liver (Lv et al., 2007), adipose (Mauney et al., 2007), and bone tissues (Mathieu et al., 2006) have also been reported.

Jiang et al. (Jiang et al., 2010) functionalized chitosan/PLGA by heparin immobilization with controlled loading efficiency. One of the main benefits of introducing chitosan into PLGA microspheres is that chitosan imparts functionality due to its reactive amino groups, so that biomolecules such as heparin could be attached (Jiang et al., 2006). The compressive strength and modulus remained in the range of human trabecular bone after the heparinization process. More importantly, heparinized chitosan/PLGA scaffolds with a low heparin loading (1.71 g/scaffold) showed a stimulatory effect on cell differentiation, as indicated by enhanced osteocalcin expression as compared with a non-heparinized chitosan/PLGA scaffold. Based on these results, Jiang et al. (Jiang et al., 2006) continued to evaluate the novel scaffolds for bone regeneration in vivo. In the rabbit ulnar critical-sized-defect model created, successful bridging of the critical-sized defect on the sides both adjacent to and away from the radius occurred using chitosan/PLGA-based scaffolds. However, the addition of chitosan to PLGA led to somewhat higher inflammation and lower mineralization than for the PLGA counterpart, which is a major problem that remains to be solved.

Three-dimensional (3D) electrospun fibrous scaffolds have been suggested as a potential tissue engineering tool for bone regeneration. Shim et al. (Shim et al., 2010) thus reported a 3D microfibrous PLLA scaffold fabricated using electrospinning techniques with a subsequent mechanical expansion process. The use of these 3D scaffolds for the proliferation of osteoblasts was examined. The 3D scaffolds led to a 1.8-fold higher level of osteoblast proliferation than generally achieved for electrospun 2D nanofibrous membranes. In vivo results further showed that 3D electrospun microfibrous matrices provided a favorable substrate for cell infiltration and bone formation after 2 and 4 weeks when using a rabbit calvarial defect model.

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3D printing technology has rapidly expanded in the tissue engineering field since it was first developed at the Massachusetts Institute of Technology. Ge et al. (Ge et al., 2009) developed 3D-printed poly(lactic acid-co-glycolic acid) (PLGA) scaffolds which could support the proliferation and osteogenic differentiation of osteoblasts. Based on their in vitro study, they also evaluated PLGA scaffolds for bone regeneration within a rabbit model (Ge et al., 2009). In both the intra-periosteum and the iliac bone defect models, the implanted scaffolds facilitated new bone tissue formation and maturation over a time period of 24 weeks.

The current clinical use of PLA-based scaffolds nevertheless remains very limited (Iwasa et al., 2009), mainly because of the risk of disease transmission and immune response. This can be illustrated by taking cartilage tissue engineering as an example. Traditional autologous chondrocyte implantation (ACI), first introduced by Brittberg et al. in 1994 (Brittberg et al., 1994), has yielded good clinical results (Bentley et al., 2003). To date, none of the short- or mid-term clinical and histological results using scaffolds were reported to be better than ACI. As for the scaffolds, collagen and hyaluronan-based matrices are among the most popular scaffolds in clinical use nowadays, since they offer substrates which are normally essential elements in native articular cartilage (Iwasa et al., 2009). Among the very few cases of scaffolds in clinical use is the copolymer of PGA/PLA (polyglactin, vicryl) and polydioxanone, which is used for cartilage repair under the trade name of BioSeed®-B and BioSeed®-C (Biotissue Technologies AG, Freiburg, Germany).

In summary, tissue engineering is one of the most exciting interdisciplinary fields today and is growing rapidly with time. The inclusive criteria for studies on scaffolds capable of clinical application were in vivo or clinical studies and thus certain artificially designed scaffold features (such as pore size, interpore connectivity, etc.) are necessary for optimal tissue engineering applications (accelerated tissue regeneration). Suggestions for future directions include the use of designer scaffolds with in vivo experimentation, and coupling scaffold design with cell printing to create designer material/biofactor hybrids to optimize tissue engineering treatments (Hollister, 2005).

5.2 Delivery systems

There has long been a desire to achieve the targeted delivery of bioactive compounds to areas in the body to maximize therapeutic potential and minimize side-effects. Many types of particles have been tested as delivery tools for biomedical applications such as liposomes, solid lipid nanoparticles, and biodegradable polyesters like PLA and PLGA (Torchilin, 2006). With its excellent biocompatibility, biodegradability, mechanical strength, heat processability, and solubility in organic solvents, PLA can be used to produce dosage forms such as pellets, microcapsules, microparticles (MP), nanoparticles (NP), etc. MP and NP of PLA, modified or unmodified, are increasingly investigated for sustained release and targeted drug, peptide/protein, and RNA/DNA delivery applications because of their small size enabling their permeation through biological barriers such as the blood-brain barrier (Roney et al., 2007).

Although PLA-based materials such as PLGA have been FDA-approved and are clinically available, they lack chemical functionalities to facilitate specific cell interactions. Furthermore, their potential for the sustained release of hydrophilic molecules (e.g. proteins) is often limited (Fahmy et al., 2005). Frequent undesired effects include low

encapsulation efficiency and high burst release of the encapsulated biomolecule within the first few hours or days, which is mainly due to the desorption of surface-associated hydrophilic molecules having weak interactions with the polymer (Fahmy et al., 2005). To circumvent these limitations and establish therapeutic efficacy, large doses or site-specific administration are often required for devices comprised of polyester biomaterials. In an attempt to address these problems, numerous groups have introduced functional groups (such as amine functionalities) on these materials, either through direct conjugation or device fabrication with additives (Betram et al., 2009).

As for drug release from MPs or NPs, it is generally controlled by both drug diffusion and polymer degradation. To ensure the efficacy of drug delivery, control over the particle size and particle size distribution is critical, since smaller particles and narrower size distributions facilitate the design of targeted drug delivery systems. These involve binding fragments specific to a tumor-associated surface antigen, with a ligand binding to its corresponding receptor on the tumor cell surface, which can be attached on the surface of the PLA-based materials. Furthermore, polymers that display a physicochemical response to changes in their environment are being intensively explored as potential drug and gene delivery systems. The use of stimuli-responsive nanocarriers offers an attractive opportunity for targeted delivery, in which the delivery system becomes an active participant rather than a passive vehicle. The advantage of stimuli-responsive nanocarriers becomes obvious when the stimuli are unique to disease pathology, allowing the nanocarrier to respond specifically to the pathological characteristics. For instance, in solid tumors, the extracellular pH decreases significantly from 7.4 (the pH value under normal physiological conditions) to about 6.5 (Vaupel et al., 1989; Haag, 2004). In addition, the pH ranges from 4.5 in lysosomes to about 8.0 in mitochondria. Given these pH shifts, therapeutic compounds with a pK_a between 5.0 and 8.0 are able to exhibit dramatic changes in physicochemical properties (Ganta et al., 2008). Another option is thermo-sensitive polymeric micelles, containing a hydrophobic core and a thermosensitive shell, the later changing from a hydrophilic nature at body temperature to a collapsed hydrophobic polymer at a hyperthermic condition of 42°C (Na et al., 2006). Investigations concerned with this theme include responses induced by chemical substances, changes in temperature (Tyagi et al., 2004) or pH (Sethuraman & Bae, 2007), electric signals (Sawahata et al., 1990) or other environmental conditions (Qiu & Park, 2001).

The use of nucleic acids as therapeutic agents for genetic diseases has been extensively studied (Torchilin, 2008). However, a major limitation of this technique lies in the low delivery efficiency of the therapeutic DNA to the diseased site. To address this issue, various strategies have been explored including vectors engineered from viruses (Brun et al., 2008) and PLGA in NP formulations. PLGA NP have shown particular promise in improving the delivery efficacy (Kocbek et al., 2007). Besides, the physical characteristics of the nanoparticles can be manipulated to escape the degradative endosomal lumen, resulting in cytosolic localization. To develop novel administration paths, hybrid versions of research have been conducted on this subject, yet the results are mostly based on animal models or in vitro results, making it difficult to draw final conclusions. From clinical trials, substantial obstacles to their use, such as immunogenicity and inflammatory potential, have also been demonstrated (Nafee et al., 2007). Therefore, there is still a long way to go before real clinical applications come through.

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Some examples of delivery systems incorporating PLA are provided in Table 5 and in Fig. 5 (Chen et al., 2007; Sethuraman & Bae, 2007). Sethuraman et al. (Sethuraman & Bae, 2007) developed a novel drug delivery system for acidic tumors consisting of two components: 1) A polymeric micelle with a hydrophobic core of PLLA and a hydrophilic shell of PEG conjugated to TAT (a cell-penetrating peptide in HIV), and 2) a highly pH-sensitive copolymer of poly(methacryloyl sulfadimethoxine) (PSD) and PEG (PSD-b-PEG). The final carrier, which was able to shield the micelles and expose them at slightly more acidic tumor pH levels, was achieved by complexing PSD with the TAT of the micelles. The results obtained showed significantly higher uptake of TAT micelles at pH 6.6 in comparison with pH 7.4, and that TAT not only translocated into the cells but it was also traced to the surface of the nucleus [see Fig. 5].



Fig. 5. Test results for PLA-based drug delivery materials. Fluorescent microscopy images are shown on top for COS7 cells transfected by plasmid encoding enhanced green fluorescence protein (EGFP) with different carriers: (A) lipofectamine, (B) methoxypolyethyleneglycol-PLA-chitosan nanoparticles (MePEG-PLA-CS NP); the transfection efficiency, as detected by flow cytometry, is higher in (B) than in (A) (Reproduced with the permission from Chen, J. et al. (2007). Preparation, characterization and transfection efficiency of cationic PEGylated PLA nanoparticles as gene delivery systems, Journal of Biotechnology, Vol.130, No.2, pp.107. Copyright (2007) Elsevier) At the bottom are dual label confocal micrographs for MCF-7 cells incubated with TAT micelles: (a) Cells stained with fluorescein isothiocyanate (FITC) attached to TAT in the micelles; (b) the same nuclei as in (a) were stained with TOPRO-3; (c) superimposed images of (a) and (b); the yellow color shows the localization of TAT within the nuclei (Reproduced with the permission from Sethuraman, V. A. & Bae, Y. H. (2007). TAT peptide-based micelle system for potential active targeting of anti-cancer agents to acidic solid tumors. Journal of Controlled Release, Vol.118, No.2, pp.216. Copyright (2007) Elsevier).

| Material | Application | Results | Reference |
|-----------------|---------------------|-------------------------------|--------------|
| PLA-PEG | Carrier for tetanus | Enhanced transport across | Vila |
| particles | toxoid | the rat nasal mucosa | et al., 2005 |
| PEG-PLA NP | Conjugated with | Increased uptake of the Lf- | Hu |
| | lactoferrin (Lf) | NP by bEnd.3 cells | et al., 2009 |
| PLA-b-Pluronic- | Carrier for oral | Good control over blood | Xiong |
| b-PLA | insulin | glucose concentration | et al., 2007 |
| PLA NP | Carrier for HIV | Induced seric and mucosal | Aline |
| | p24 proteins | antibody production | et al., 2009 |
| Surfactant-free | Carrier for HIV | Elicited strong CTL response | Liggins |
| PLA NP | p24 proteins | and cytokine release | et al., 2004 |
| PLA | Carrier for | Reduced inflammation of | Jie |
| microspheres | paclitaxel | arthritis rabbit model | et al., 2005 |
| PEO-PLA | Carrier for 5-FU | Complete drug release | Zhang & |
| copolymers | and paclitaxel | | Feng, 2006 |
| PLA-TPGS | Carrier for | Initial burst followed by | Freitas |
| copolymers | paclitaxel | sustained release | et al., 2005 |
| PLA | Carrier for | Initial burst followed by an | Chen |
| microspheres | nimesulide | exponential decrease | et al., 2007 |
| PEGylated PLA | Gene delivery | Improved transfection | Ataman-Önal |
| NP | systems | activity | et al., 2006 |
| PLA-PEG-PLA | Carrier for 5-FU | Good control over the release | Venkatraman |
| copolymer | and paclitaxel | | et al., 2005 |
| AP-PEG- | Drug carrier for | Presented high tumor- | Wu |
| PLA/MPEG-PAE | cancer therapy | specific targeting ability | et al., 2010 |
| PLGA/PEI NP | Carrier for | Effective silencing of the | Patil & |
| | luciferase siRNA | gene in cells | Panyam, 2009 |
| cNGR-PEG-PLA | Carrier for DNA | Rapid and efficient | Liu |
| NP | | nanoparticle internalization | et al., 2011 |
| DMAB coated | Loaded with | Significantly improved | Fay |
| PLGA NP | plasmid DNA | transfection efficiencies | et al., 2010 |

Table 5. Investigations on PLA-based material as drug delivery systems. AP: peptide, CRKRLDRN; MPEG: methyl ether poly(ethylene glycol); PAE: poly(β-amino ester); PEI: polyethylenimine; cNGR: Cyclic Asn-Gly-Arg; DMAB: dimethyldidodecylammonium bromide.

In summary, some problems still remain to be tackled for this promising novela dministration method. A major issue is the presence of surfactants such as SDS or stabilizers such as PVA in the microparticles, necessary to achieve antigen binding and colloidal stability. Although present only at low concentrations, the acceptability of such components in human vaccines depends on the results of extensive and costly toxicological studies. Biodegradable polymers used for drug delivery to date have mostly been in the form of injectable microspheres or implant systems requiring complicated fabrication processes with organic solvents. In such systems, the organic solvents can denature components such as protein drugs being encapsulated. Besides, these delivery systems have relatively low transfection efficiencies in vitro as compared with reagents commercialized for cell transfection. The last problem concerns the lack of test results for these delivery systems

using animal models or in clinical trials, which is of fundamental importance for real applications in biomedical therapy.

5.3 Other fields

Due to its versatility, PLA has been investigated for membrane applications (e.g. wound covers), implants and medical devices (fixation rods, plates, pins, screws, sutures, etc.), and dermatological treatments (e.g. facial lipoatrophy and scar rejuvenation).

With respect to wound treatment, bacterial infections are one of the main factors impacting the healing process. One of the best approaches to treat wound infections is by the immobilization of drugs or antibacterial agents within the nanofibers by electrospinning, or the electrospinning of polymers with intrinsic antibacterial and wound-healing properties. Dozens of patents have been issued on that topic so far (Ghosh et al., 2007; Robinson et al., 2009). Silver nanoparticles (nAg) and the natural polysaccharide chitosan (as well as its quaternized derivatives) are most commonly used as antibacterial agents with a high intrinsic activity against a broad spectrum of bacteria (Rujitanaroj et al., 2008; Ignatova et al., 2009).

Metals are still the most popular materials for fracture fixation, but their disadvantages include stress shielding, accumulation in tissues, hypersensitivity, growth restriction, pain, corrosion, and interference with imaging techniques. Consequently, the focus of research is increasingly shifted to biomaterials like PLA, which offers satisfactory strength during the healing of bone tissue and then degrades over time (Mavrogenis et al., 2009). A commercial product with a proven track record in clinical applications is the VICRYLTM suture material, based on PGA/PLA copolymers (Mehta et al., 2005). The number of applications of PLA as fixation rods, plates, pins, screws, sutures, etc. in orthopaedics and dentistry is also increasing (Raghoebar et al., 2006).



Fig. 6. Applications of PLA in the biomedical field.

In 2004, Sculptra[™] [poly(*L*-lactic acid)] was approved by the FDA as the first injectable facial "volumizer" in the treatment of lipoatrophy due to its significant therapeutic effectiveness (Burgess & Quiroga, 2005). The lipodystrophy syndrome is associated with the usage of highly active antiretroviral therapy (HAART) containing protease inhibitors or nucleoside reverse transcriptase inhibitors for HIV patients. The action mechanism of Sculptra[™] is via stimulation of the fibroblastic activity with generation of collagen and other connective tissue fibers. In addition, it acts as dermal matrix adding support by thickening the dermis (Vleggaar & Bauer, 2004). Moreover, PLA can help improve the appearance of scars due to acne, surgery, trauma, or suture (Lowe & Beer, 2005, as cited in Beer & Rendon, 2006).

6. Conclusions

Due to the multiple desirable characteristics of PLA including renewability, biocompatibility, transparency, and thermoplasticity, it is being used or is a potential candidate for many consumer and biomedical applications (Jamshidian et al., 2010). Ever increasing environmental concerns associated with conventional polymers derived from petrochemicals lead to constantly expanding applications for PLA since its discovery in 1932 by Carothers at DuPont.

In previous years, the most negative point of PLA was its higher price as compared with petrochemical-based polymers. Today, by optimizing the LA and PLA production processes, and with increasing PLA demand, a reduction in its price can be achieved. The price of PLA is currently much lower than in previous years. Meanwhile, PLA is mainly synthesized in the industry by ROP employing tin(II) bis(2-ethylhexanoate) (SnOct₂) as a catalyst, which has been approved as a food additive by the FDA, but the potential toxicity associated with most tin compounds cannot be ignored for biomedical applications. Scientists all over the world are now exploring novel, well-defined catalysts with good biocompatibility, high catalytic activity, low toxicity, and excellent stereoselectivity. This should remain an everlasting interest area. Finally, the possibility of obtaining 100% biosourced opens the way for PLA to become more independent from petrochemical-based polymers, free of environmental and health concerns.

However, the major disadvantages of PLA such as its poor ductility, slow degradation rate, and poor hydrophilicity somewhat limit its applications. The modification of PLA bulk and surface properties has thus become crucial to increase its applicability. Many of the bulk and surface modification strategies discussed above have been designed to tune the PLA surface properties according to the demands of biomedical applications. Unfortunately, all these established methods for surface modification are somewhat deficient and while they provide control over the wettability, degradation rate, and functionality, it is still compulsory to minimize their negative impact on PLA bulk properties. Thus a combined modification strategy (e.g. irradiation followed by grafting) or a better balance of PLA surface and bulk properties should be sought. Ideally, with respect to a better balance of properties and shorter modification times, one-step approaches need to be developed because it is time-consuming to carry out surface and bulk modifications separately, and the solvents and reagents involved in multiple modification steps tend to affect PLA properties significantly.

All these modification strategies aim at tailoring the properties of PLA-based materials for certain applications. Fortunately, more and more encouraging results have been reported,

but the present conclusions from most of these reports cannot be directly generalized to truly biomedical applications since most of the experiments were carried out in vitro. Nevertheless, these findings offer some clues for further improvements. The increasing number of functional PLA-based polyesters provides the opportunity to study the relationships between structure and functionality of these polymers such as cell adhesion and degradability in vitro and in vivo, as well as to develop applications of these materials for delivery system in the form of micro- and nanoparticles or scaffolds for tissue engineering. Finally, cancers and acquired or inherited genetic diseases represent one of the most serious threats to the health of human beings, but no effective therapies are available so far. It is suggested that the development of effective and ideal tools for drug, peptide/protein, and RNA/DNA delivery will represent a good alternative in drug development. It therefore appears that it would be best to focus future research work on the rational design of novel carriers for biomedical uses and targeted delivery systems. Obviously, this requires plenty of relevant experiments on animal model and enough clinical trials before they are widely utilized.

Even though countless studies have focused on the synthesis and the modification of PLA and remarkable progresses has been achieved over the last two decades, vast opportunities as well as challenges remain in terms of exploring the characteristics of PLA-based materials and expanding their domains of applications.

7. Acknowledgement

It was supported by the Natural Science of Hubei Province for Distinguished Young Scholars (No. 2008CDB279) and the Key Technologies R & D Program of Hubei Province (20091933), the Fundamental Research Funds for the Central Universities, Huazhong University of Science and Technology (2010JC016) and the National High Technology Research and Development Program ("863" Program) of China (No. 2008AA10Z339). The authors are also grateful to Dr. Wen-bin Zhang from the University of Akron for his valuable suggestions during the preparation of this manuscript.

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Biomedical Science, Engineering and Technology

Edited by Prof. Dhanjoo N. Ghista

ISBN 978-953-307-471-9 Hard cover, 902 pages **Publisher** InTech **Published online** 20, January, 2012 **Published in print edition** January, 2012

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How to reference

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Lin Xiao, Bo Wang, Guang Yang and Mario Gauthier (2012). Poly(Lactic Acid)-Based Biomaterials: Synthesis, Modification and Applications, Biomedical Science, Engineering and Technology, Prof. Dhanjoo N. Ghista (Ed.), ISBN: 978-953-307-471-9, InTech, Available from: http://www.intechopen.com/books/biomedical-science-engineering-and-technology/poly-lactic-acid-based-biomaterials-synthesis-modification-and-applications



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